

Simulation of Viscoplastic Deformation of Low Carbon Steel Structures at Elevated Temperatures

Y. Sun, K. Maciejewski, and H. Ghonem

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The deformation response of a low carbon structural steel subjected to high temperature simulating fire conditions is generated using a viscoplastic material constitutive model which acknowledges the evolution of the material hardening parameters during the loading history. The material model is implemented in an ABAQUS subroutine (UMAT) which requires the determination of the material constants as a function of temperature. Both the temperature dependency and strain-rate sensitivity of the material parameters have been examined by the analysis of a single steel beam and a steel-framed structure subjected to temperatures ranging from 300 to 700 °C. Sequentially coupled thermal-stress analysis is applied to a structure under simulated fire condition. Results of this analysis show that above a transitional temperature, the deformation of the steel is strain-rate dependent. The combined effect of heat flux and loading rate on the complex deformation of a two-story steel structure is examined and the significance of employing a viscoplastic material model is discussed.

Keywords finite element analysis, hardening, internal state variables, strain-rate sensitivity, structural steel, viscoplasticity

1. Introduction

With the advancement of computation technology, engineers tend to adopt more advanced thermal and mechanical simulation techniques to assess the safety and reliability of engineering structures. Advanced numerical analysis allows less simplifications or assumptions for the description of material constitutive behavior, geometric modeling, real boundary conditions and loading, hence leading to a more realistic prediction of the stress/deformation in the studied engineering structure. For example, the deformation response prediction of steel-framed structures under fire conditions was done based on an elasto-plastic beam-column formulation (Ref 1, 2); the non-uniform profile of temperature across section of frame members was considered in Ref 3; the creep strain model was included in the simulation of steel frames in fire (Ref 4); nonlinear analyses for three-dimensional steel frames in fire was presented in Ref 5, 6, but without taking into account the strain-rate sensitivity of the steel material at elevated temperature and nonlinear temperature distribution both across section of steel frames and along length of beam column.

In this study, viscoplastic constitutive equations (Ref 7, 8) are used to describe the nonlinear behavior of low carbon steel at elevated temperatures. The material parameters in the viscoplastic constitutive equations are modeled as a function of temperature and implemented in ABAQUS subroutine UMAT.

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The UMAT code is verified by the simulation of tensile tests of monotonic displacement loading at different temperatures and of a fatigue test of cyclic displacement loading. The simulation of single steel beam is conducted to show the effect of loading rate and temperature on the beam deflection. Furthermore, sequentially coupled thermal-stress simulation technique is employed in the finite element analysis of a three-dimensional steel-framed structure. Transient heat transfer analysis is first conducted to obtain the temperature distribution on the steel frames as a function of time. Nodal temperature results from heat transfer analysis are then transferred for structural response analysis. The main focus of this article is to study the influence of the mechanical behaviors of the steel material on the deformation response of steel structures.

2. Viscoplastic Constitutive Equations

The model described in this section is based on unified constitutive equations in the manner developed by Chaboche and Rousselier (Ref 7, 9). This model formulated on the assumption that a viscoplastic potential, Ω , exists in the stress space. A particular form of the viscoplastic potential is (Ref 8):

$$\Omega = \frac{K}{\alpha(n+1)} \exp\left(\alpha \left\langle \frac{\sigma_v}{K} \right\rangle^{n+1}\right), \quad (\text{Eq 1})$$

where K , n , and α are material parameters that characterize the rate sensitivity. To describe the viscoplastic behavior, the concept of time-dependent overstress or viscous stress, σ_v , is used. This is given as:

$$\sigma_v = J(\sigma - X) - R - k, \quad (\text{Eq 2})$$

where the tensor X represents the kinematic hardening stress, R and k are scalar variables corresponding to the isotropic hardening stress and the initial yield stress, respectively, and $J(\sigma - X)$, is Von Mises second invariant defined by:

$$J(\sigma_{ij} - X_{ij}) = \left(\frac{3}{2} (\sigma'_{ij} - X'_{ij}) : (\sigma'_{ij} - X'_{ij}) \right)^{1/2} \quad (\text{Eq 3})$$

where σ' and X' are the deviatoric parts of the applied stress tensor, σ , and the back stress tensor, X , respectively. The scalar k is a temperature-dependent material constant representing the initial size of the elastic domain. Throughout this development, the total strain is partitioned into an elastic component and an inelastic component. The unified viscoplastic equations incorporate the plastic and creep components simultaneously in an inelastic or viscoplastic strain component denoted by ε_p throughout the rest of this article (Ref 7, 9, 10). The relation between plastic flow and the viscoplastic potential is determined by means of the normality rule:

$$d\varepsilon_p = \frac{\partial \Omega}{\partial \sigma_{ij}} = dp \frac{3}{2} \frac{(\sigma'_{ij} - X'_{ij})}{J_2(\sigma'_{ij} - X'_{ij})}, \quad (\text{Eq 4})$$

where the dp is the accumulated plastic strain rate, which is given in terms of effective plastic strain as:

$$dp = \left(\frac{2}{3} d\varepsilon_p d\varepsilon_p \right)^{1/2} = \left\langle \frac{J_2(\sigma - X) - R - k}{K} \right\rangle^n \times \exp \left(\alpha \left\langle \frac{J_2(\sigma - X) - R - k}{K} \right\rangle^{n+1} \right) \quad (\text{Eq 5})$$

The kinematic hardening terms, measured by the back stress X , describes the internal changes during each inelastic transient. Two terms of back stress describing the phenomena of the Bauschinger effect will be considered here. The first term, X_{ij}^1 , is the short-range hardening effect, which is a fast saturated variable. The second term, X_{ij}^2 , is a quasi-linear variable describing the long-range hardening effect. Hence, the back stress is defined as:

$$X_{ij} = X_{ij}^1 + X_{ij}^2 \quad (\text{Eq 6})$$

where each term is described by a general form as (Ref 7, 8):

$$\dot{X}_{ij}^i = c_i \left(\frac{2}{3} a_i \dot{\varepsilon}_{ij}^p - X_{ij}^i \dot{p} \right) - \beta_i \left| J(X_{ij}^i) \right|^{r_i-1} X_{ij}^i \quad (\text{Eq 7})$$

The first term describes the strain hardening as a function of X , plastic strain rate and accumulated plastic strain rate. The second term represents processes for bypassing or penetrating barriers at comparable rates and is termed dynamic recovery. The third term, which is strongly temperature dependent, describes the time-recovery or static-recovery as a function of X . c , α , β , and r are material-dependent parameters and $J(X_{ij})$ is the second invariant of X_{ij} .

The slow evolution of microstructure associated with cyclic hardening or softening of the material can be described by the isotropic hardening variable, R , which is the difference in the saturation position after a loading cycle and that corresponding to the monotonic loading for the same plastic strain. This is governed by the following equations:

$$\dot{R} = b(Q - R)\dot{p} \quad (\text{Eq 8})$$

$$Q = Q_{\max}(1 - e^{-\mu q}) \quad (\text{Eq 9})$$

$$q = \max(|\varepsilon_p|, q) \quad (\text{Eq 10})$$

where Q and b are the limiting values of the isotropic hardening variable. Q is the saturation limit of R while the constant,

b , is a temperature and material-dependent parameter describing how fast R reaches Q . This latter parameter can be either a positive value, indicating cyclic hardening, or a negative value, indicating cyclic softening. Q_{\max} is the maximum value of Q , and q is the maximum strain achieved during loading, which memorizes the previous plastic strain range (Ref 11).

3. Model's Material Parameters

The material employed in this study is A572 Grade 50 Low Carbon Steel and its microstructure which is shown in Fig. 1, includes pearlite colonies and α -ferrite equiaxed grains. The average grain size is 53 μm (ASTM 5). Volume fraction of the pearlite phase is 10%.

A series of strain-controlled tests were carried out on the as-received steel, at both room and high temperature, to determine the various material parameters described above in order to fully identify the nonlinear kinematic hardening model. The mechanical testing was carried out using a servo hydraulic test machine, equipped with a heat induction coil for the high temperature tests. The strain was measured with a quartz rod extensometer.

A monotonic test is carried out, to determine the modulus, E , and yield stress, k , of the material. Results of this rate at various loading conditions are shown in Fig. 2(a). A series of strain-controlled fully reversed cyclic stress-strain tests (fatigue stress ratio = -1) are performed at strain ranges varying from $\pm 0.2\%$ to $\pm 1\%$ strain. A typical cyclic stress-strain curve at 300 $^\circ\text{C}$ is shown in Fig. 2(b). These loops are employed to generate the isotropic and kinematic hardening, as well as, viscosity and recovery parameters.

The isotropic hardening parameter, b , is determined from the evolution of the peaks of the cyclic stress-strain curves (Fig. 2b) Q_{\max} and μ are calculated from the difference in the peak stress of the first cycle and the saturated cycle as a function of the maximum plastic strain achieved during loading (Ref 8, 11–13). The kinematic stress corresponds to the center of the linear part of the first reversible cyclic loop at each strain

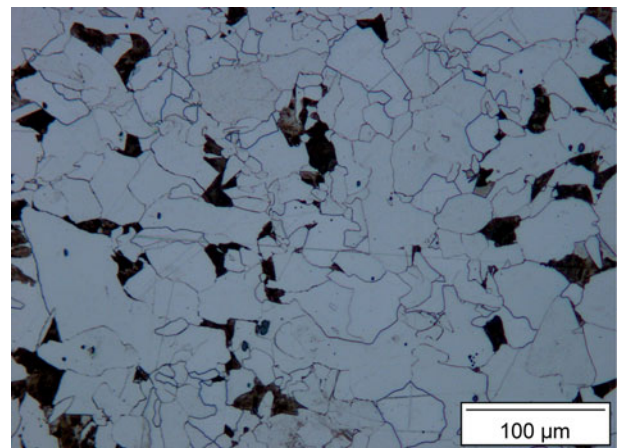


Fig. 1 Optical micrograph of as-received A572 Grade 50 steel etched for 5 s in 5 vol.% nital. The microstructure is typical of normalized steel, consisting of pearlite colonies (dark phase) and α -ferrite (light phase) equiaxed grains. The pearlite colonies have a volume fraction of 10% and the grain size is 53 μm (ASTM 5)

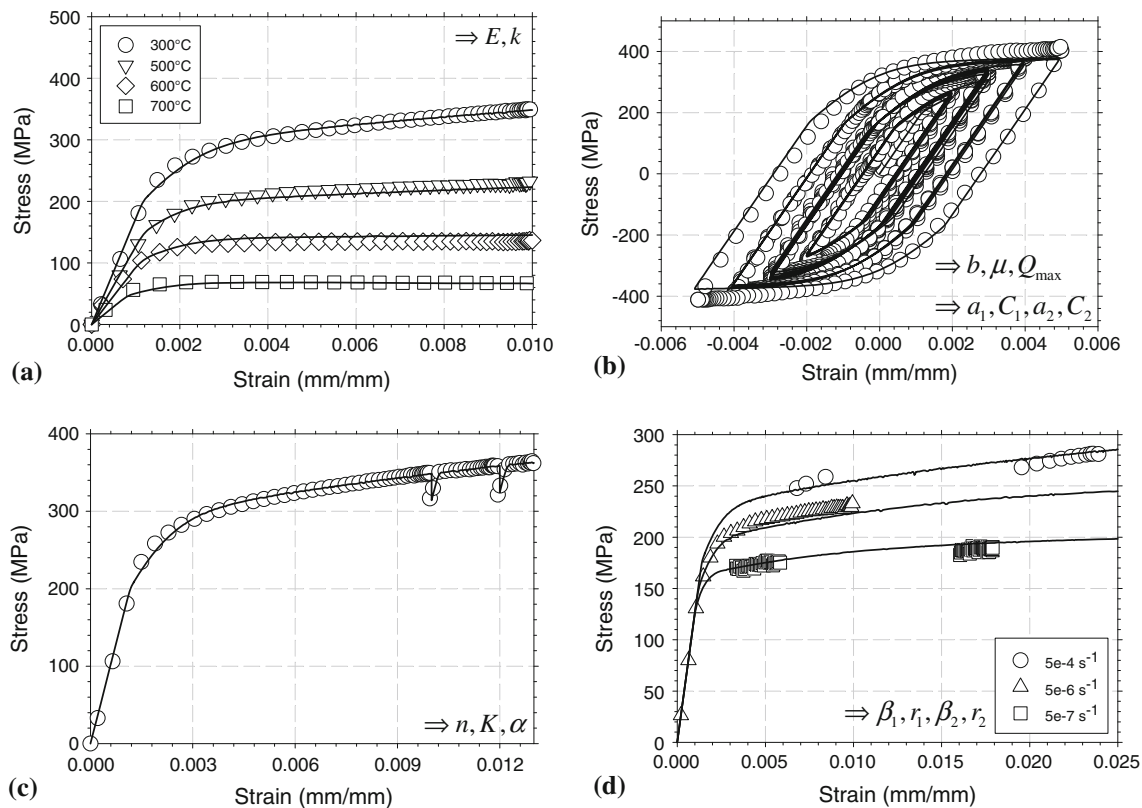


Fig. 2 Experimental (symbol) and numerical (solid line) stress-strain curves for low carbon steel tested. (a) Monotonic stress-strain curves, (b) cyclic stress-strain curves at 300 °C, (c) relaxation stress-strain curves at 300 °C, and (d) monotonic stress-strain curves at multiple strain rates at 500 °C. These curves are used to determine the material parameters specified on the graphs

range (Fig. 2b). This stress is determined as a function of plastic strain to determine the materials parameters a_1 , c_1 , a_2 , and c_2 .

To describe the time-dependent viscous stress term, a strain-controlled monotonic stress-relaxation test is performed, as seen in Fig. 2(c). For this, during the periods of holding at a constant total strain, the stress and stress rate as a function of time is acquired. This is used to obtain the viscous stress in terms of plastic strain rate, from which, n and K , strain-rate sensitive parameters must be determined (Ref 8, 10, 14). The material constant, α , is taken to be the saturation limit of viscous stress for high-plastic strain rates.

Time-dependent recovery parameters, β_1 , β_2 , are determined from the linear portion of the stress time curve. In this region it is assumed that R , k and viscous stress are constant and the time-dependent stress is only coming from the time-dependent back stress. These parameters, β_1 , β_2 , r_1 and r_2 , are optimized with stress-strain data at various strain rates. Strain-rate sensitivity tests (Ref 14) are performed in which a specimen is loaded monotonically in strain control at multiple strain rates as shown in Fig. 2(d).

The material parameter determination procedure, described above, has been applied to as-received steel tested at temperatures ranging from 20 to 700 °C, and is further detailed in Ref 8, 10–14. The material constants from the above procedure are shown in Table 1 for each temperature condition.

Several of the material constants, as described above, are temperature dependent. The material constants have been determined for five different temperatures, ranging from room

temperature to 700 °C. A linear interpolation to the material constants is generally used to extend a constitutive model working for any temperature, which may be less accurate. A nonlinear form better describes the temperature span, and the formula used to fit the temperature-dependent material constants is given below:

$$C = A * \tanh\left(-\left(\frac{T}{T_0}\right)^m\right) + C_0, \quad (\text{Eq 11})$$

where C and T represent a material constant and temperature, respectively, A , m , T_0 , and C_0 are unknown parameters which are figured out by curve fitting to $C - T$ data.

Figure 3 shows two examples of the curve fitting to the temperature-dependent Young's modulus and strain-rate sensitivity coefficient using the Eq 11.

The saturation of material constants at lower temperature was captured by the hyperbolic tangent function of temperature. The saturation values are defined by the parameters, A and C_0 . The sensitivity of the material constant change with temperature is defined by the exponent, m . The reference temperature, T_0 , defines the temperature at the middle of the transitional segment of the curve. The parameters for temperature-dependent material constants are listed in Table 2.

The temperature-dependent and independent material constants are used to fully define the viscoplastic constitutive model. This model is then implemented in finite element software to describe the material behavior under varying loading and temperature conditions.

Table 1 Material parameters for as-received A572 Grade 50 low carbon steel tested at temperatures ranging from 20 to 700 °C

Temperature	20 °C	300 °C	500 °C	600 °C	700 °C
E , GPa	188	172	127	93	58
k , MPa	100	71	27	11	3
a_1 , MPa	137	96	54	36	22
C_1	1411	1411	1411	1411	1411
a_2 , MPa	1864	1753	1111	566	175
C_2	1.15	1.15	1.15	1.15	1.15
b	78	78	78	78	78
Q_{\max} , MPa	86	78	31	1	-10
μ	1447	1447	1447	1447	1447
n	14	14	14	14	14
K , MPa	155	349	304	227	107
α	$5.5E+06$	$5.0E+06$	$1.0E+05$	$4.5E+00$	$9.9E-01$
β_1 , Pa ^{2.58}	$1.0E-21$	$9.9E-20$	$1.0E-15$	$1.0E-15$	$1.0E-15$
β_2 , Pa ^{2.15}	$4.6E-26$	$2.1E-16$	$1.0E-12$	$1.0E-12$	$1.0E-12$
r_1	2.58	2.58	2.58	2.58	2.58
r_2	2.15	2.15	2.15	2.15	2.15

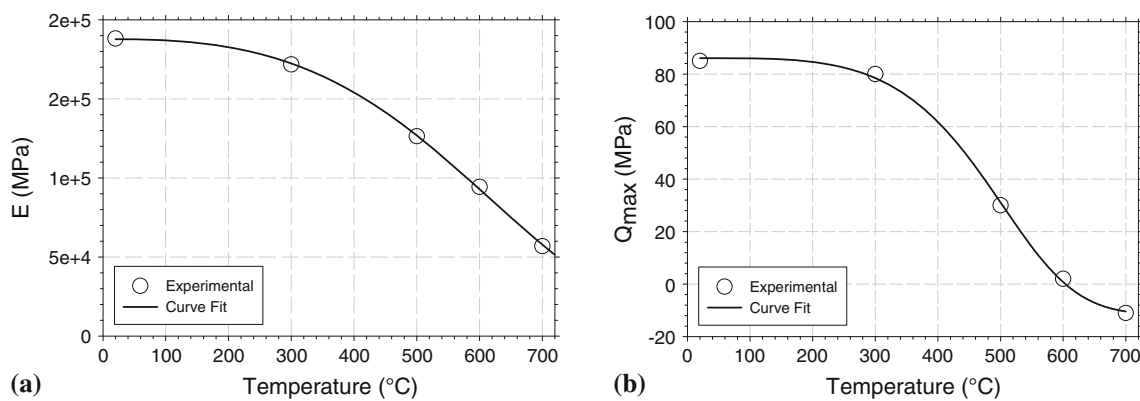


Fig. 3 Modeling of temperature-dependent material constants. (a) Young's modulus and (b) isotropic hardening coefficient, Q_{\max}

Table 2 Parameters in Eq 11 for temperature-dependent material constants

Constant	A	T_0	C_0	m
E	187850.1	742.0	187743.8	2.8
k	100.6	520.8	99.6	2.3
a_1	137.4	621.3	137.5	1.6
a_2	1830.5	619.1	1864.3	3.9
Q_{\max}	97.7	558.0	86.1	4.1
K	351.7	720.1	351.5	5.4
α	$5.5E+06$	437.3	$5.5E+06$	6.4
β_1	$-1.0E-15$	443.0	$2.7E-27$	18.8
β_2	$-1.0E-12$	443.0	$3.2E-24$	18.8

4. Validation of UMAT Code

The viscoplastic constitutive equations have been implemented into ABAQUS through a subroutine UMAT. A finite element model of uniaxial fatigue specimen was built to simulate the experimental tensile and fatigue tests. A comparison of experimental and numerical stress-strain curves is used to examine the implementation of the simulation model. A quarter 3D finite element model is shown in Fig. 4. The

diameter of the cylindrical specimen is 6.35 mm and the gage length is 10 mm. The block element with eight nodes was used to mesh the geometry model. Nodes on the bottom plane were fixed in vertical direction, and symmetry boundary conditions were applied to the nodes on two side planes. To simulate the tensile tests, nodes on the top plane were subjected to a displacement ramping from 0 to 0.1 mm within 2000 s. Four simulation cases were done by assigning four different uniform temperatures to the whole model. The predicted stress-strain curves are shown in Fig. 5 and compared with the experimental results which show good agreement.

The quarter cylinder model was also used for the simulation of a fatigue test at 300 °C. A cyclic displacement loading (see Fig. 6a) was applied to the nodes on the top plane.

The magnitude of applied displacement equals the value that the gage length of the model times the strain that the element was subjected to. The displacement loading shows three strain ranges, $\pm 0.2\%$, $\pm 0.3\%$, and $\pm 0.4\%$. Four cycles were run under strain range $\pm 0.2\%$, six cycles under strain range $\pm 0.3\%$, and three cycles under strain range $\pm 0.4\%$. The cyclic test of three strain ranges used the same loading strain rate of 5×10^{-6} mm/mm/s. The period of one cycle under strain range $\pm 0.2\%$ is 1600, 2400 s under strain range $\pm 0.3\%$, and 3200 s under strain range $\pm 0.4\%$. The element stresses in vertical direction were output and plotted with the strains in the

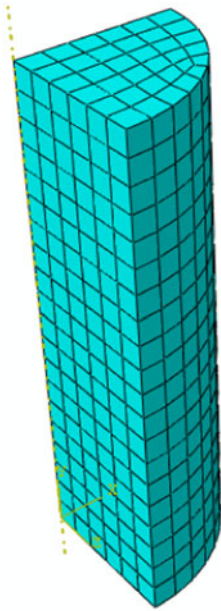


Fig. 4 FE model for tensile tests

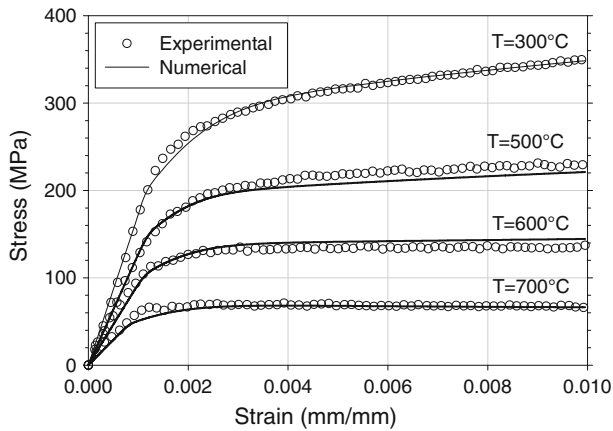


Fig. 5 Numerical and experimental stress-strain curves

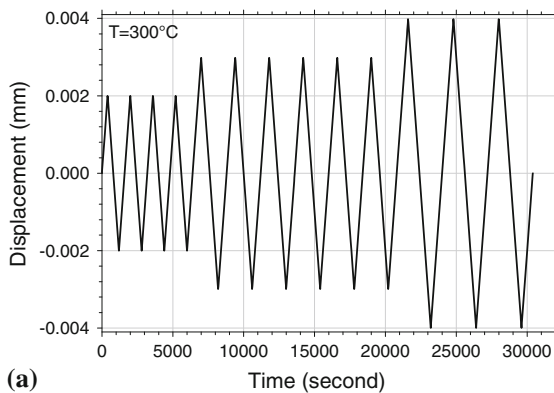


Fig. 6 (a) Cyclic strain loading in fatigue tests and (b) experimental and numerical hysteresis stress-strain curves at 300 °C

same direction. Both experimental and numerical hysteresis stress-strain curves are shown in Fig. 6(b). The good agreement of experimental and numerical stress-strain curves validates the implementation of the UMAT code.

5. Finite Element Analysis and Results

5.1 Simulation of Single Steel Beam

Numerical simulations were conducted using two finite element models, a simple model of single steel beam and a complex model of steel-framed structure. Simulation of single steel beam model (Fig. 7) was built to examine effects of loading rate and temperature on the beam deflection.

The two-dimensional beam is 3 m long and 0.2 m high. The element type of plain strain was assigned to all elements. Each element has a size of 0.1 m long, 0.02 m high. Uniform temperature field was applied to the beam and two temperature cases, 300 and 700 °C, were studied. Uniform pressure ramping up with time was also applied on the beam. Two cases of pressure ramping rate were investigated; 1.5 and 0.15 MPa/min.

The beam deflections from the four simulation cases described above are compared in Fig. 8, which shows the evolution of the maximum beam deflections along with the ramping pressure.

Figure 8(a) compares the beam deflections generated at 300 °C for two different pressure ramping rates, while Fig. 8(b) shows the comparison at 700 °C. At the temperature 300 °C, the beam deflection is not dependent on the loading rate; however, it shows dependency at the temperature 700 °C. This result indicates that the 300 °C is below the transition temperature of the material strain-rate sensitivity. If we compare the beam deflections under the same pressure load but at the two different temperatures, it becomes obvious that the beam deflection is larger at 700 °C. The higher temperature reduces the strength of the material steel or the stiffness of the steel beam.

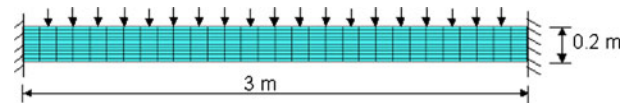
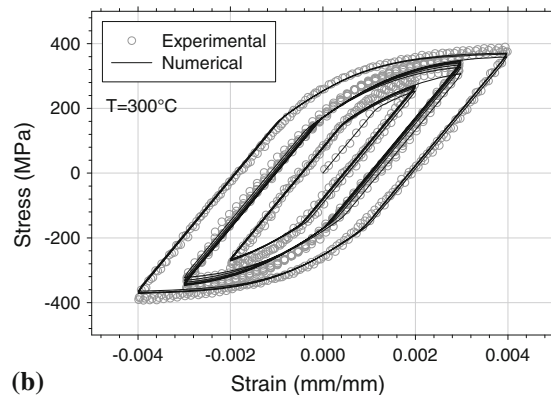


Fig. 7 Single steel beam model



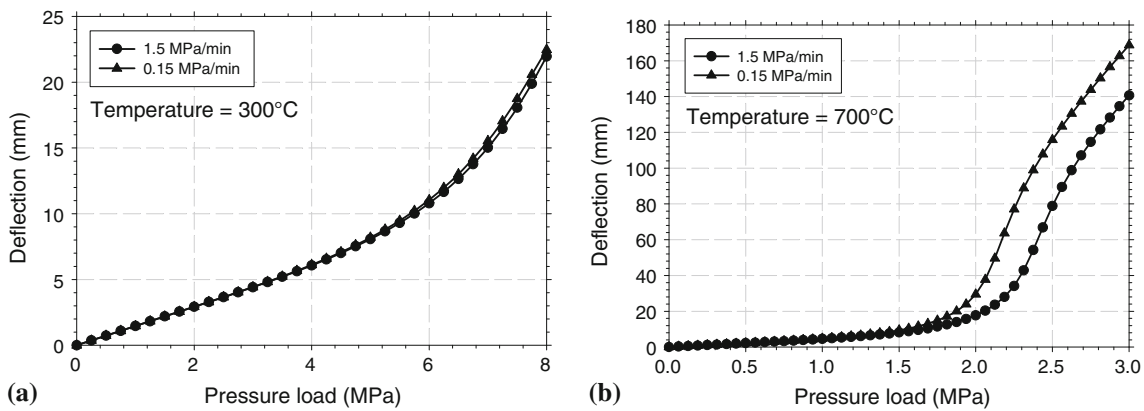


Fig. 8 Beam deflection at two loading rates (a) at temperature 300 °C and (b) at temperature 700 °C

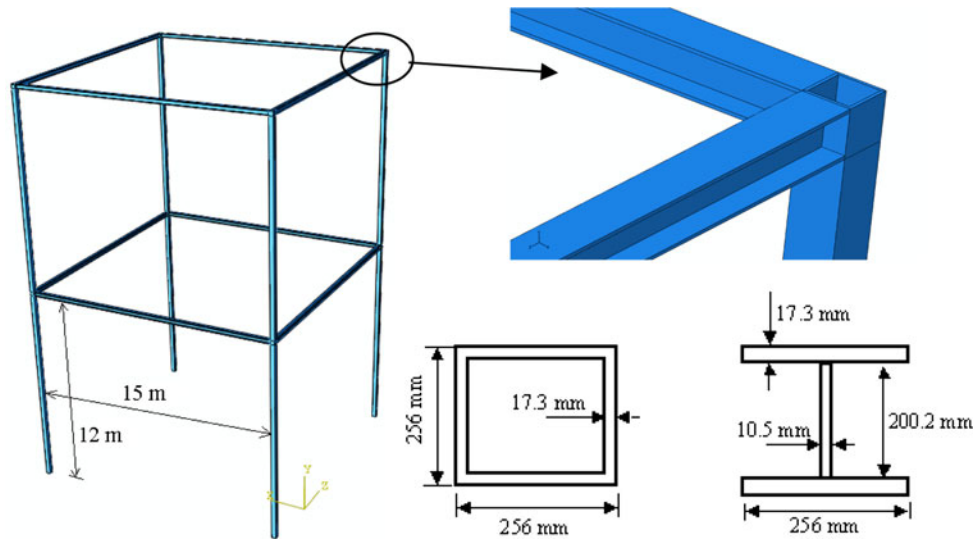


Fig. 9 FE model of a 3D steel-framed structure under fire condition

5.2 Simulation of Steel-Framed Structure Under Fire Condition

Finite element analysis of steel-framed structures under fire condition uses a sequentially coupled thermal-stress analysis (Ref 15). In this analysis, a transient heat transfer analysis is first performed to obtain temperature distribution history, and then a stress/deformation analysis is conducted to obtain structural deformation. Nodal temperatures are calculated in the transient heat transfer analysis and stored as a function of time in the heat transfer result file. The stress analysis uses the same geometry model with the same meshing as the heat transfer analysis. The temperature fields for the stress analysis are coupled with temperature results from transient heat transfer analysis.

A three-dimensional model of a multi-story structure was built, the dimensions of which are shown in Fig. 9.

The structure has two cells with eight box-shape columns and eight I-shape beams. Each column is 12 m high and each beam is 15 m long. The dimensions of the column and beam cross sections are shown in Fig. 9. We consider the beams and columns to be integral. Eight-node block elements were used to mesh the structure as shown in Fig. 10.

The element size is 0.5 m in longitude direction of beam and column, and <0.04 m in their transverse direction. The total number of elements is 19,136.

5.2.1 Heat Transfer Analysis. The finite element model was first employed to conduct transient heat analysis. The material properties required for heat transfer analysis are shown in Table 3.

The element type was set as DC3D8 for heat transfer analysis. The room temperature 25 °C was assigned to the steel-frame structure as the initial condition. A surface heat flux of 16 kW/m² was applied on the bottom plane of one beam in the first floor, and 24 kW/m² on the side plane of its connected column in order to simulate the appearance of a fire in the cell (see Fig. 11).

The heat transfer duration is 1 h. The calculated temperatures at all nodes and at different time are stored in the output file of heat transfer analysis. The output frequency of nodal temperatures is 0.2 Hz.

Figure 12 shows the temperature distribution after 1 h of heat transfer. The non-uniform temperatures across the sections of steel frames were observed.

The higher temperature occurred to the positions near the surface where external heat flux was applied. Within 1 h, the

heat spread was confined in the mid cell of the first floor. Positions far away from the surface where the heat flux came in almost maintained their initial temperature of 25 °C. Figure 13 and 14 show the temperature history at seven points on the heated beam's and column's cross sections, respectively.

The temperature at the monitored positions increases with time since the continuous heat flow increase the thermal energy of the structure. For the beam, the temperatures at points B1 and B2 are higher than those at points B3, B4, and B5, and much higher than those at points B6 and B7. For the column, the temperatures at points C1 and C2 are higher than those at points C3, C4, and C5, and much higher than those at C6 and C7. Although B1 and B2 are on the same heat flux surface, their temperatures are different. The temperature at B1 is higher than that at B2 because the position B2 is nearer to the web, which is the heat transfer path.

5.2.2 Stress/Deformation Analysis. In the stress/deformation analysis, the element type was changed from heat transfer type to stress analysis type C3D8, while the geometry

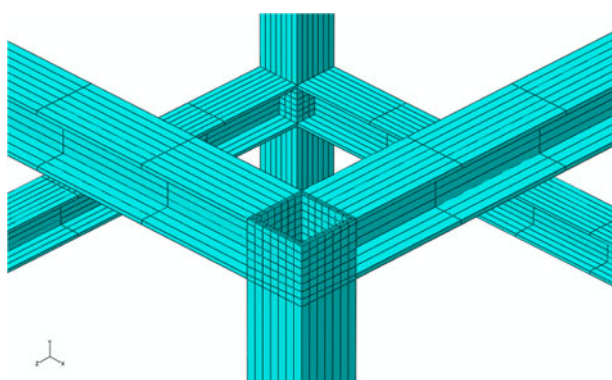


Fig. 10 Meshing of FE model of 3D steel-framed structure

Table 3 Material properties for heat transfer analysis

Mass density	Thermal conductivity	Specific heat
7800 kg/m ³	50 J/m ² °C	470 W/kg °C

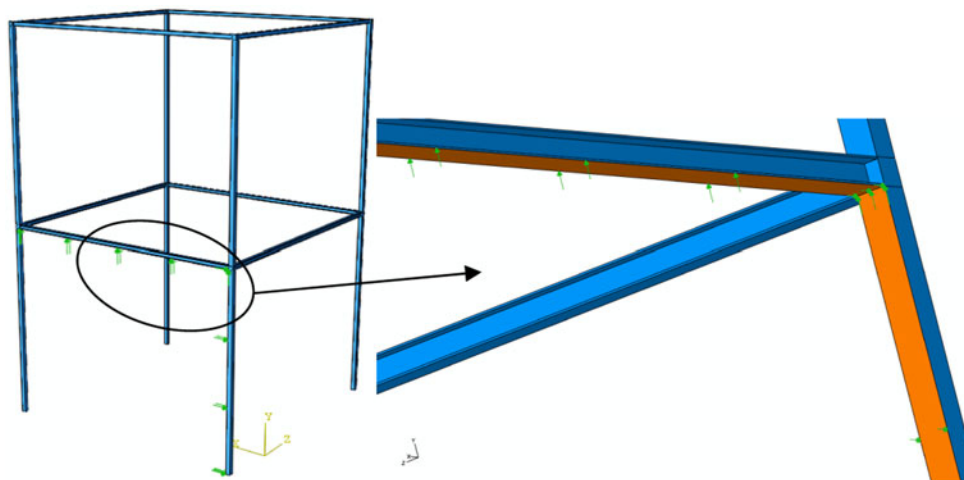


Fig. 11 Surface heat flux as thermal loading applied to the structure

remained the same. The developed UMAT was used as the material model. The four columns were fixed at their roots. A uniform pressure was applied on all beams and temperature field was assigned to the whole structure. The simulation consists of three loading steps. The first step is to increase the uniform pressure from 0 to 0.04 MPa in 10 s, while keeping the temperature constant at 25 °C. In the second step, the maximum pressure loading is maintained and the temperature field is coupled with the results obtained from the transient heat transfer analysis, and hence the duration of step 2 is 60 min. In the third step, both the pressure and temperature distribution in the structure is held constant for 30 min. Therefore, static stress/deformation analysis was performed with a time period of 90 min plus 10 s. The maximum time increment is 0.5 s. The control setting for the nonlinear effect of large displacement was activated in the simulation. The displacement, strain, and stress results along with time were output from the stress analysis.

The final deformation of the three-dimensional two-story structure from the stress analysis using the viscoplastic constitutive model, discussed above, was examined. Among the four beams of the first floor, the beam subjected to the heat flux deflected more than other beams as expected. Figure 15 shows the deflection history of two beams at the first floor (see Fig. 10).

The heated beam refers to the beam which had the heat flux applied to it, and the non-heated beam is a beam neighboring the heated beam. In the 10 s of load step 1, the deflections of the two beams increase quickly due to the increasing pressure load. In this step, the two beams achieved the same amount of deflection due to geometrical symmetry, the same loading condition, and uniform structural temperature. In load step 2, the deflection of the heated beam does not increase at the beginning although its temperatures increase. It is concluded that the beam's deflection starts to increase when the applied load/stress becomes larger than the strength of the beam. The applied load did not change but the beam's strength degraded with the increase of temperature. When the strength of the heated beam degraded below its stress level, the inelastic strain occurred as shown in Fig. 16, and the beam's deflection started to increase slowly initially and then faster with the continuously increasing of the beam's temperature.

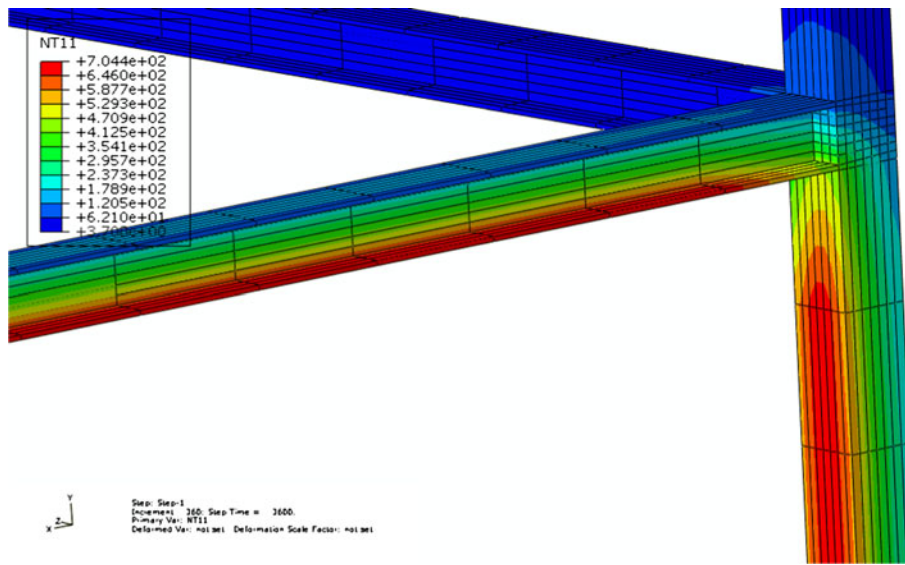


Fig. 12 The final temperature distribution in the structure

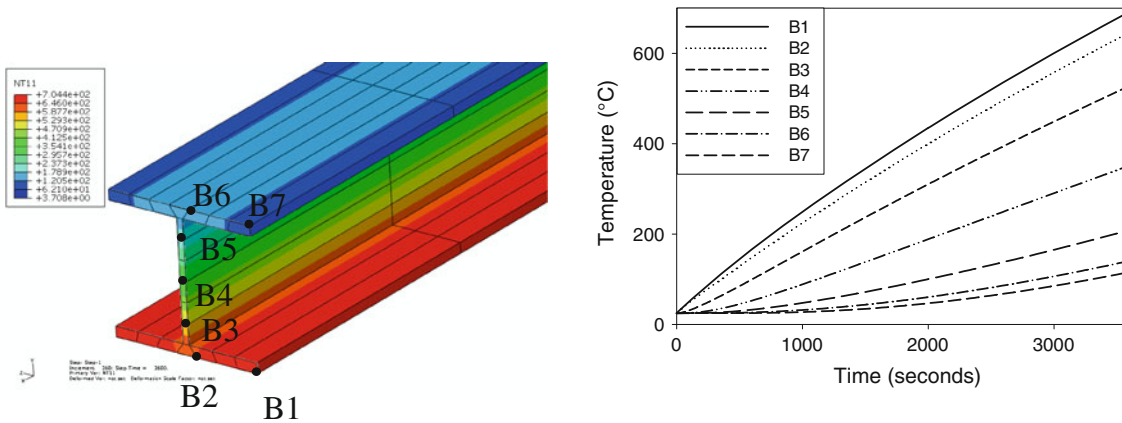


Fig. 13 Temperature history of seven points on the beam's cross section

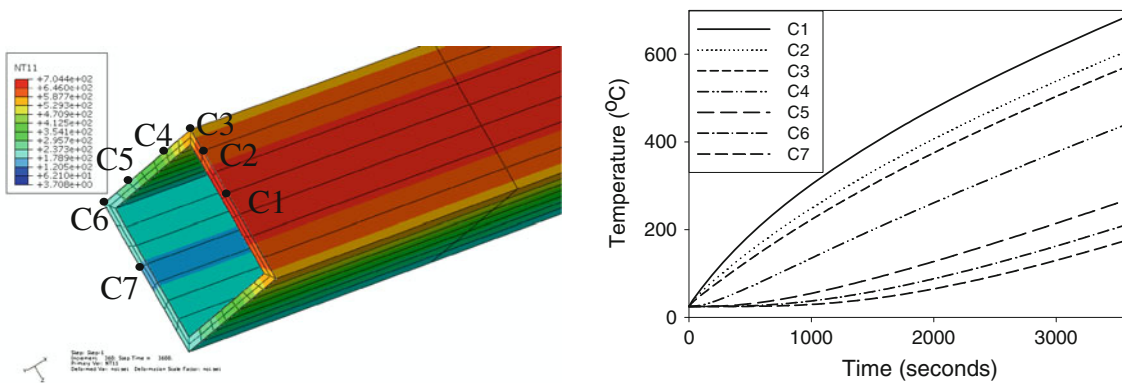


Fig. 14 Temperature history of seven points on the column's cross section

In load step 3, the applied loading and the temperature distribution did not change (actually the applied stress relaxed as shown in Fig. 16), but the heated beam's deflection continued to

increase. The deflection increase in load step 3 was attributed to the creep deformation (that is, material strain-rate sensitivity) at elevated temperatures, which can be shown in Fig. 16.

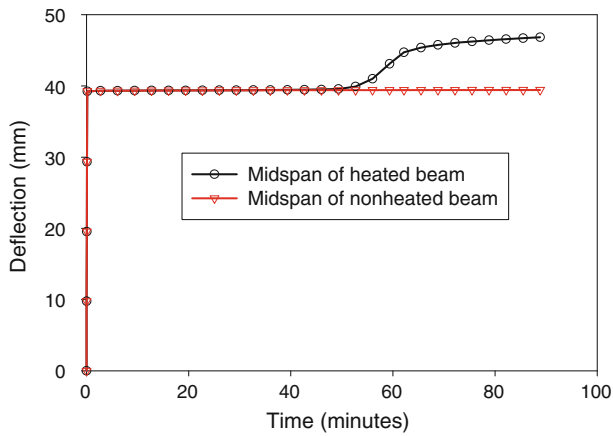


Fig. 15 The deflection history at the midspan of the heated and non-heated beams

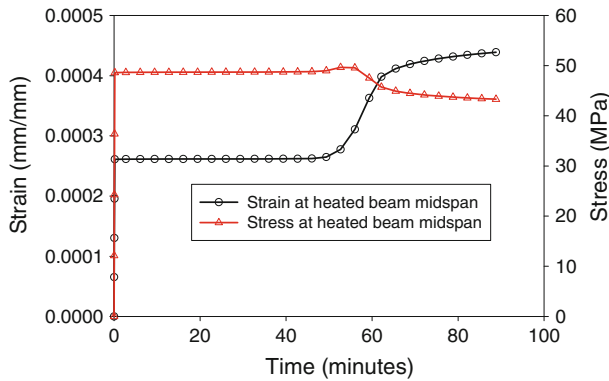


Fig. 16 Strain and stress history at the midspan of the heated beam

6. Conclusions

The work in this article describes a viscoplastic constitutive model that has been employed to simulate the flow behavior of structural steel A572 Grade 50 steel at elevated temperatures. The material constants of the model have been fitted as a function of temperature, thus allowing the implementation of the corresponding constitutive equations in a UMAT subroutine of the ABAQUS platform. The validation of the UMAT has been carried out by comparing the experimental data with those obtained from the numerical simulations under the same applied load conditions. The finite element analysis of a three-dimensional steel-framed structure under fire condition has been performed using a simulation strategy of sequentially coupled thermal-stress analysis. The general conclusions of this study can be listed as follows:

1. The nonlinear material behavior of the low carbon structural steel at elevated temperatures can be captured by unified constitutive equations, which consist of a nonlinear kinematic hardening model and a hyperbolic function for the modeling of temperature-dependent material constants.

2. The steel shows a transition temperature for the time-dependent plastic deformation. The strain-rate sensitivity transition is indicated by the trend of temperature-dependent material constants.
3. The heat transfer analysis of the 3D steel-frame structure shows the temperature distributed nonlinearly across the sections of the steel frame and along the length of beams and columns.
4. As the temperature increases, large bending or buckling deformation could occur to the steel-frame structure as a result of the degradation of the material's strength below the applied stress level.
5. The time-dependent material behavior contributes to the structure's bending or buckling when the applied temperature is higher than the strain-rate sensitivity transition temperature.

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